

# Transport of styrene monomer through natural rubber

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The diffusion and transport of inhibitor-free styrene through crosslinked natural rubber (NR) have been studied at various temperatures. NR has been vulcanized by conventional, efficient, peroxide and mixed vulcanization techniques. The dependence of diffusion coefficient on the crosslinking system has been studied for all the systems. The influence of temperature on the sorption and the activation energies of sorption have been calculated. The interaction parameter, permeability, sorption coefficient and molecular weight between crosslinks have been evaluated using the diffusion data. The effect of degree of crosslinking on the sorption characteristics of styrene through NR has also been investigated for the different crosslinking systems. The peroxide system showed lowest uptake and the conventional system showed highest uptake.

(Keywords: crosslinked natural rubber; diffusion; sorption)

## INTRODUCTION

The diffusion properties of different liquids in polymers have been studied extensively by several authors<sup>1–6</sup>. Often the diffusion of liquids through rubbery polymers can be described by Fick's law of diffusion. However, there are instances where deviation from Fick's law has been observed. The transport phenomena of various solvents and monomers through polymers have also been investigated. The sorption properties of rubbery polymers are controlled by polymer structure, crosslink density, type of crosslinking, temperature, etc. In the study on swelling characteristics of natural rubber (NR) in aromatic hydrocarbon liquids, Harogoppad *et al.*<sup>7</sup> investigated the dependence of transport phenomena on polymer structure. The effect of crosslink density on diffusion has been observed by Poh *et al.*<sup>8</sup>. The dependents of swelling and absorption of given samples on temperature have also been investigated<sup>9–11</sup>. Harogoppad *et al.* made use of the temperature dependence of the transport coefficients to estimate the activation parameters from the Arrhenius plots<sup>1</sup>.

The swelling and sorption behaviours of NR in various liquids are of great interest, as NR is a well known polymer having extensive use in everyday life. The swelling of natural rubber samples in the presence of monomers like styrene, methyl methacrylate, acrylonitrile, butyl acrylate, etc., is of great research interest, since it constitutes the preliminary step in the preparation of interpenetrating polymer networks (IPN)<sup>12–14</sup>. To prepare a rubber-based IPN, the swelling character-

istics of the rubber in monomers are to be studied in detail. A clear and accurate knowledge about the swelling rate of the polymer in the monomers helps in controlling the composition, morphology and properties of the resulting IPN. However, no detailed investigation has been made so far on the kinetics of diffusion and sorption of styrene through natural rubber crosslinked by different curing systems such as conventional (CV), efficient (EV), peroxide (DCP) and mixed systems.

In this paper the swelling of NR in styrene monomer has been studied in the temperature range of 25–45°C. This will be the primary step for the development of IPN based on NR and polystyrene (PS) having various morphological features and property profiles. The NR was vulcanized by different techniques, viz. conventional, efficient, peroxide and mixed vulcanization systems. The influence of different vulcanizing systems on diffusion properties is analysed. For each crosslinking system, samples were vulcanized at different cure times so as to attain samples of different crosslink densities. The effect of crosslink density on sorption has been investigated. Finally the temperature dependence of diffusion and rubber-solvent interaction parameter  $\chi$  are determined.

## EXPERIMENTAL

### Materials

The natural rubber (NR) used was of ISNR-5 grade and was supplied by Rubber Research Institute of India (RRII). The styrene monomer is made free of inhibitor by washing with 0.1% NaOH and dried using sodium sulfate before use for diffusion experiments. The chemicals used in the various vulcanizing systems are of laboratory reagent grade.

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### Sample preparation

The formulations for different mixes are given in Table 1. The mixing was done in a two-roll mixing mill, as per ASTM procedure. The rheographs are taken for each mix to obtain the cure times (Figure 1). The mixes are cured at different time intervals:  $t_{100}$ ,  $t_{90}$ ,  $t_{80}$ . From the rheograph, the time corresponding to the maximum torque was taken as  $t_{100}$ . The optimum cure time,  $t_{90}$ , refers to the time for obtaining 90% of the maximum torque, i.e. 90% of  $t_{100}$ . The time corresponding to 80% of maximum torque was taken as  $t_{80}$ . The cure characteristics of the samples are given in Table 2.

### Immersion/weight-gain experiments

For the swelling experiments, circular samples are cut from the different vulcanized sheets using a sharp steel die (diameter = 1.9 cm). The thickness of the sample was

measured by means of a screw gauge. The samples are weighed on an electronic balance and then kept immersed in inhibitor-free styrene monomer in test bottles. The samples are taken out of the liquid at specific time intervals, the monomer adhering to the surface rubbed off, and the samples weighed and replaced in the test bottles. This process was continued till equilibrium is reached (i.e. after about 50 h). The time for each weighing was kept to a minimum (at most 40 s) to make the error due to the escape of solvent from the sample negligible. The values obtained are found to be perfectly reproducible. When conducting the experiments at temperatures higher than room temperature, the samples were kept to a thermostatically controlled air oven.

The mole per cent uptake,  $Q_t$ , for the monomer by 100 g of the polymer was plotted against the square root of time and the results are analysed<sup>15</sup>. The  $Q_t$  value is defined by:

$$Q_t = \frac{M_e(m)/M_r(m)}{M_i(s)} \times 100 \quad (1)$$

where  $M_e(m)$  is the mass of monomer at equilibrium,  $M_r(m)$  is the relative molecular mass of the monomer, and  $M_i(s)$  is the initial mass of the sample. When equilibrium was reached,  $Q_t$  was taken as  $Q_\infty$ , i.e. mole uptake at infinite time.

## RESULTS AND DISCUSSION

To obtain conclusions from the diffusion experiments, the sorption curves were analysed. The  $Q_\infty$  values are given in Table 3. Figures 2, 3 and 4 show the diffusion curves of NR crosslinked by the four vulcanized systems having optimum cure ( $t_{90}$ ) at three different temperatures, 25, 35 and 45°C. The figures show clearly that monomer uptake is higher in the case of conventional and mixed systems and lowest in the peroxide system. The efficient vulcanizing system takes an intermediate position. The difference in the  $Q_\infty$  values for the various systems is partly due to the different types of crosslinking present in them. The structure of networks formed during vulcanization is shown in Figure 5. The polysulfidic linkages in conventional vulcanization impart high chain flexibility. In the mixed system both polysulfidic and C–C linkages are present. But as DCP is a slow curing reagent and as the cure time for the mixed system was 10 min, which is not enough to effect complete peroxide curing (the vulcanization time for the NR/DCP system is 25 min (Table 2)), the linkages

**Table 1** Formulations for the mixes

Ingredient (phr) <sup>a</sup>	CV	EV	DCP <sup>b</sup>	Mixed
NR	100	100	100	100
Stearic acid	1.5	1.5		
Zinc oxide	5	5		
MOR <sup>c</sup>	0.6			0.6
TMTD <sup>d</sup>		1		
CBS <sup>e</sup>		1.5		
DCP			4	1.5
Sulfur	2	0.6		1.5

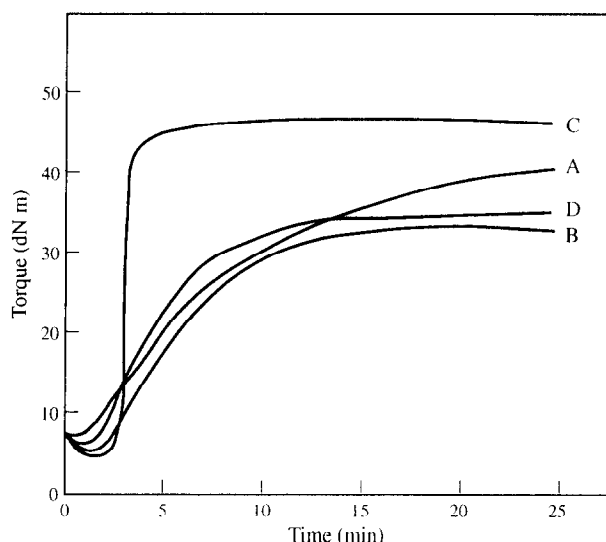
<sup>a</sup> Parts per hundred rubber by weight

<sup>b</sup> Dicumyl peroxide

<sup>c</sup> Morpholinebenzothiazyl sulfenamide

<sup>d</sup> Tetramethylthiuram disulfide

<sup>e</sup> N-Cyclohexyl-2-benzothiazyl sulfenamide



**Figure 1** Rheographs of the mixes: A, DCP; B, mixed; C, EV; D, CV

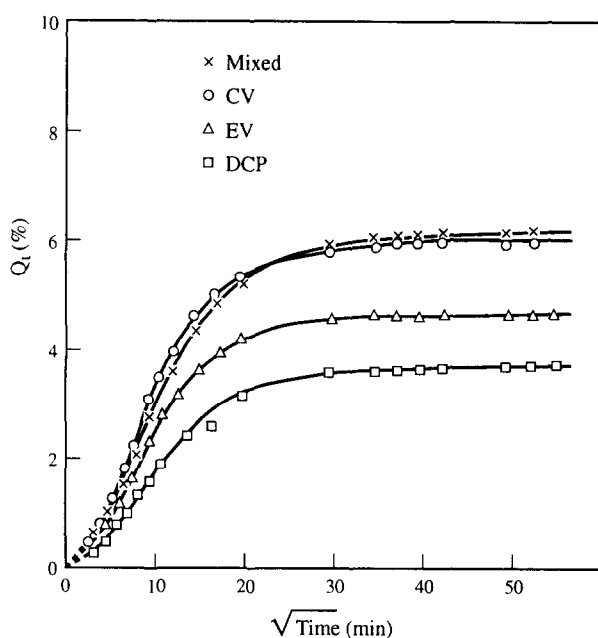
**Table 2** Cure characteristics of the mixes

Cure characteristics	CV	EV	DCP	Mixed
Minimum torque (dN m)	5.6	4.5	7	5
Maximum torque (dN m)	36	47.4	44.8	33.8
Scorch time (min)	2.0	2.5	1.8	2.4
Optimum cure time (min)	9	5	25	10

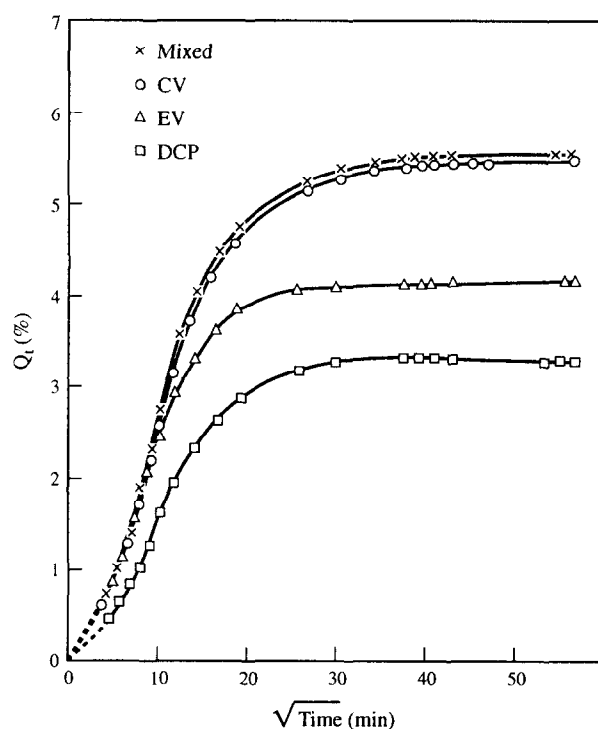
**Table 3** Values of  $Q_\infty$  for different samples

Temp. (°C)	Cure time	CV	EV	DCP	Mixed
25	$t_{100}$	6.0935	4.4989	3.3623	6.029
	$t_{90}$	5.4790	4.2800	3.3819	5.597
	$t_{80}$	5.5497	4.2273	3.4033	5.5786
35	$t_{100}$	6.016	4.4096	3.4128	6.196
	$t_{90}$	5.5679	4.342	3.5076	5.6443
	$t_{80}$	5.4518	4.2082	3.5794	5.897
45	$t_{100}$	6.5039	4.7559	3.4619	6.050
	$t_{90}$	6.1026	4.4039	3.5076	5.752
	$t_{80}$	6.076	4.5086	3.5784	5.850

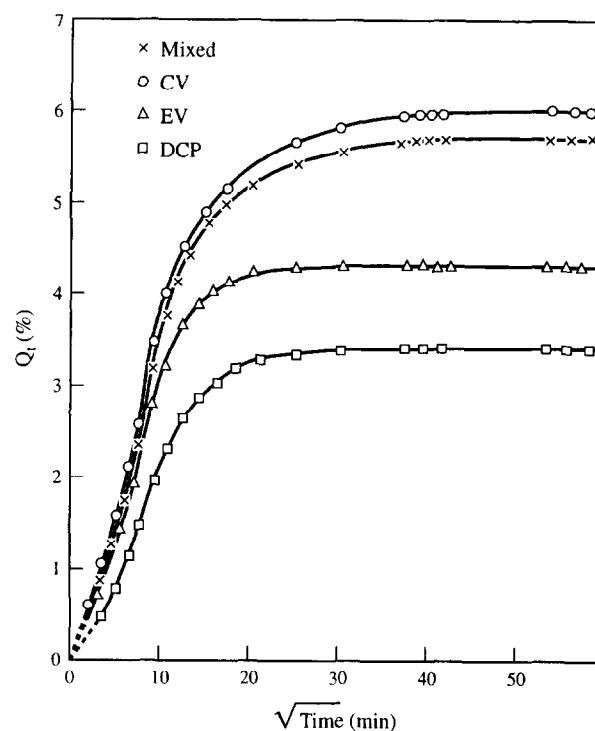
formed may be predominantly polysulfidic. As a result the monomer molecules are more readily accommodated between the rubber chains in the case of conventional as well as mixed systems. The efficiently vulcanized samples with mono- and disulfidic linkages are comparatively less flexible and monomer uptake is lower in this case. In the case of DCP-cured system, less-flexible C–C linkages are present and hence lowest  $Q_{\infty}$  values are observed. The diffusion curves for undercured and overcured samples for the four systems were also analysed. In all cases and at different temperatures the same trend was observed



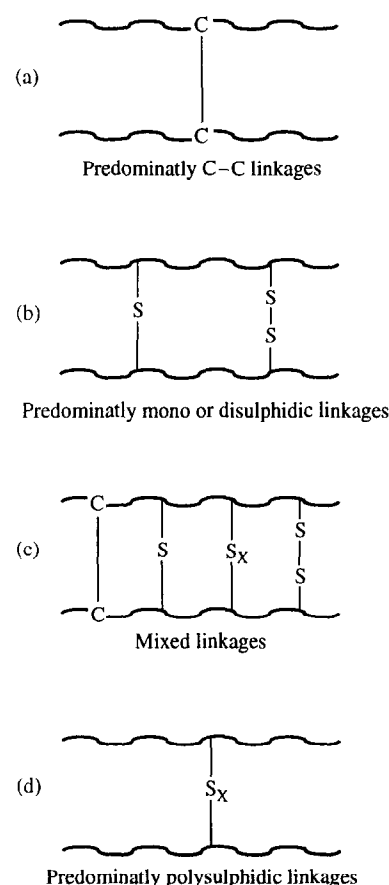
**Figure 2** Sorption curve showing the mole per cent uptake of optimum cured CV, EV, mixed and peroxide samples at 25°C



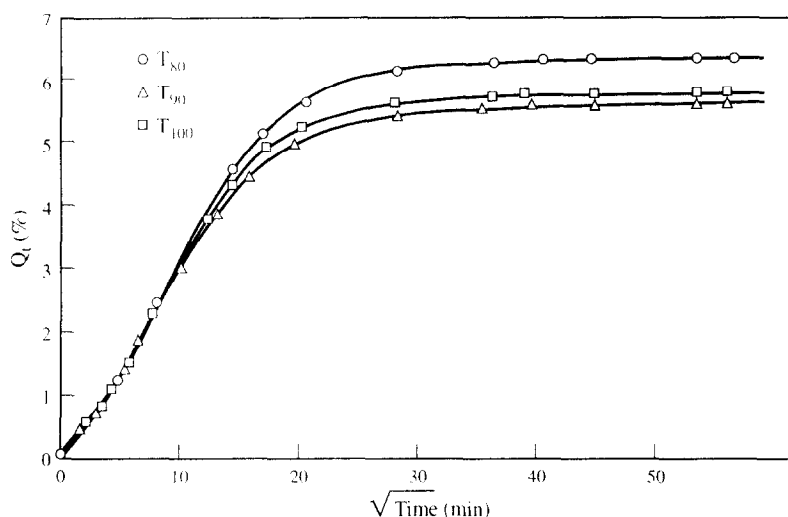
**Figure 3** Sorption curve showing the mole per cent uptake of optimum cured CV, EV, mixed and peroxide samples at 35°C



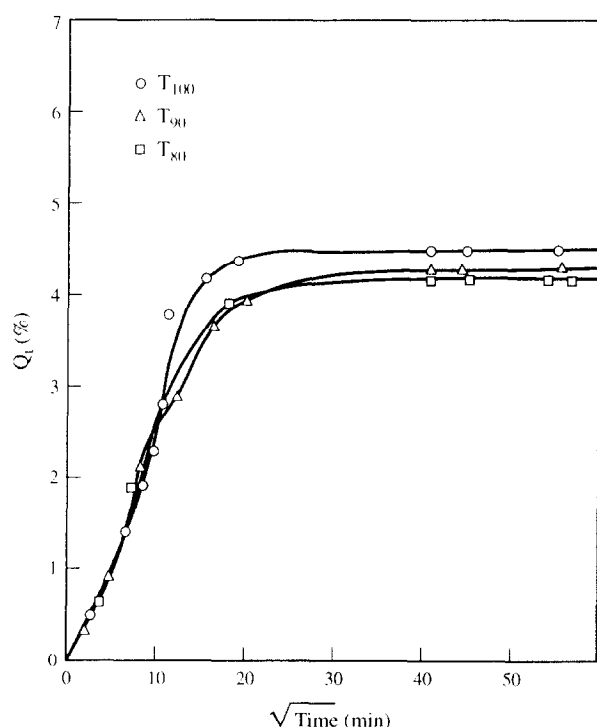
**Figure 4** Sorption curve showing the mole per cent uptake of optimum cured CV, EV, mixed and peroxide samples at 45°C



**Figure 5** Schematic representation of the crosslinks formed during curing: (a) peroxide, (b) efficient, (c) mixed and (d) conventional vulcanization



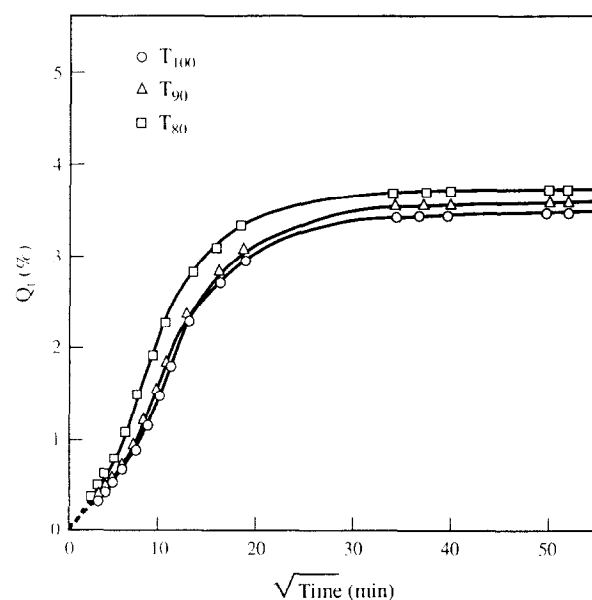
**Figure 6** Sorption curve showing the mole per cent uptake of conventional vulcanized samples with different cure times



**Figure 7** Sorption curve showing the mole per cent uptake of efficient vulcanized samples with different cure times

(CV, mixed > EV > peroxide). The other major factor that contributes to the difference in the monomer uptake is the crosslink density, which is discussed later on in this paper.

Samples cured at different cure times ( $t_{100}$ ,  $t_{90}$ ,  $t_{80}$ ) for the same vulcanizing system at the same temperature show different  $Q_{\infty}$  values, which may be explained by the degree of crosslinking. The diffusion curves for CV samples with different cure times ( $t_{100}$ ,  $t_{90}$ ,  $t_{80}$ ) at room temperature were analysed (Figure 6). The  $t_{100}$  and  $t_{80}$  samples show greater uptake than the optimum cured ( $t_{90}$ ) samples. The same is the case with efficient, peroxide and mixed systems (Figures 7, 8 and 9). The  $t_{100}$  samples are overcured and hence some of the crosslinks formed may undergo rupture. This facilitates higher monomer uptake. On the other hand, for  $t_{80}$  samples, the system is



**Figure 8** Sorption curve showing the mole per cent uptake of peroxide vulcanized samples with different cure times

undercured and therefore the crosslinks may not be fully formed. Here also monomer uptake is made easy. In the  $t_{90}$  samples the crosslinks are optimum and complete for the best balance of properties. Therefore the monomer uptake is minimum in this case. We have also analysed the behaviour at different temperatures. The trend was exactly the same. Therefore, the same explanation holds good for all samples at different temperatures.

The volume fraction of rubber,  $\phi$ , in the solvent-swollen samples was calculated using<sup>16</sup>:

$$\phi = \frac{w_1/\rho_1}{w_1/\rho_1 + w_2/\rho_2} \quad (2)$$

where  $w_1$  is the weight of the rubber sample,  $\rho_1$  the density of natural rubber,  $w_2$  the weight of styrene monomer in the swollen sample, and  $\rho_2$  the density of the monomer. The high value of  $\phi$  is an indication of high crosslinking density. The values are given in Table 4. The  $\phi$  values of DCP-cured samples were found to be higher, showing high crosslink density and hence low monomer

uptake. The  $\phi$  values are lowest for CV and mixed samples, which show high monomer uptake. For EV system the value comes in between those of CV and DCP systems, which shows the intermediate swelling characteristics. At high temperature (35 and 45°C) and

for different degrees of curing, the same pattern was followed.

The values of swelling coefficient ( $\alpha$ ) were calculated using<sup>17</sup>:

$$\alpha = \frac{M_s}{M_0} \times \frac{1}{\rho_0} \quad (3)$$

where  $M_s$  is the mass of solvent (mass of swollen sample – original mass of polymer sample),  $\rho_0$  the solvent density and  $M_0$  the mass of the polymer sample. The swelling coefficient  $\alpha$  is a measure of the volume of solvent in unit mass of the polymer. The values are given in Table 5. The swelling coefficient is in the order DCP < EV < CV, mixed. The swelling coefficient is found to increase with temperature.

To understand the mechanism of sorption, the values obtained were fitted to the equation<sup>10</sup>:

$$\log (Q_t/Q_\infty) = \log k + n \log t \quad (4)$$

where  $Q_t$  is the mole per cent increase in uptake at time  $t$ ,  $Q_\infty$  the mole per cent increase in uptake at equilibrium,  $t$  the time and  $k$  a constant characteristic of the polymer, which indicates the interaction between polymer and solvent. The values of  $n$  and  $k$  were determined by linear regression analysis. In all cases the correlation coefficients were found to be 0.99.

The value of  $n$  is equal to 0.5 for a Fickian diffusion mechanism. When the value falls between 0.5 and 1, the diffusion is anomalous. In this system, i.e. for NR and styrene, a slight deviation from the Fickian mode of diffusion is shown at all temperatures (Table 5). The values of  $k$  for different systems were also determined from equation (4) (Table 5). From the swelling data the diffusion coefficient  $D$  was calculated using<sup>6</sup>:

$$D = \pi (h\theta/4Q_\infty)^2 \quad (5)$$

where  $\theta$  is the slope of the linear portion of the sorption curves and  $h$  is the initial thickness of the sample. The values of diffusion coefficients are given in Table 6. The peroxide system shows highest  $D$  value, as it has the lowest  $Q_\infty$  values. The CV and mixed systems have lowest diffusion coefficients. Also, it is noticeable that, in each system, the value of  $D$  increases with temperature. The increase of diffusion with temperature invariably points out the activation of the diffusion process at higher temperature.

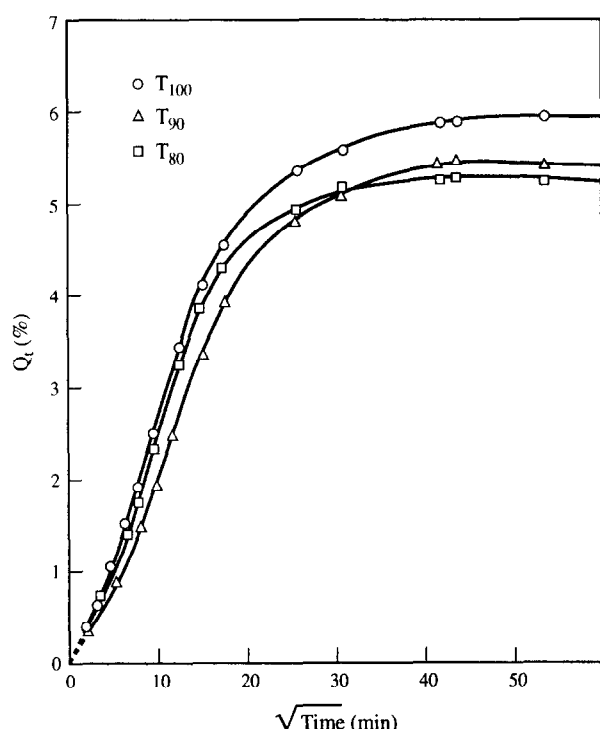


Figure 9 Sorption curve showing the mole per cent uptake of mixed vulcanized samples with different cure times

Table 4 Values of  $\phi$  for different samples

Temp. (°C)	Cure time	CV	EV	DCP	Mixed
25	$t_{100}$	0.1286	0.1666	0.2113	0.1295
	$t_{90}$	0.1411	0.1823	0.2101	0.1384
	$t_{80}$	0.1394	0.1745	0.2090	0.1401
35	$t_{100}$	0.1299	0.1693	0.2085	0.1267
	$t_{90}$	0.1415	0.1715	0.2116	0.1373
	$t_{80}$	0.1412	0.1712	0.2022	0.1322
45	$t_{100}$	0.1212	0.1588	0.2056	0.1264
	$t_{90}$	0.1284	0.1596	0.2007	0.1355
	$t_{80}$	0.1289	0.1663	0.2067	0.1334

Table 5 Values of swelling coefficient,  $n$  and  $k$  at different temperatures

Temp. (°C)	Cure time	Swelling coefficient				$n$				$k \times 10^n$ (g/g min <sup>2</sup> )			
		CV	EV	DCP	Mixed	CV	EV	DCP	Mixed	CV	EV	DCP	Mixed
25	$t_{100}$	6.771	5.123	3.855	6.913	0.6161	0.6192	0.6310	0.6627	3.05	3.19	2.66	2.12
	$t_{90}$	6.279	4.921	3.878	6.4186	0.5727	0.6344	0.5994	0.6300	3.83	3.15	2.96	2.62
	$t_{80}$	6.302	4.826	3.9029	6.3265	0.6334	0.6375	0.6537	0.5888	2.73	3.29	3.05	3.24
35	$t_{100}$	6.899	5.051	3.912	7.105	0.5452	0.5906	0.5564	0.5489	4.62	5.05	3.96	4.33
	$t_{90}$	6.252	4.979	3.8396	6.472	0.5628	0.5986	0.6456	0.6810	3.86	4.13	3.18	2.36
	$t_{80}$	6.2668	4.825	4.0119	6.763	0.6638	0.6509	0.6732	0.6418	2.76	3.33	3.34	3.05
45	$t_{100}$	7.467	8.456	3.981	7.1239	0.6075	0.5102	0.5840	0.5510	3.50	6.22	6.53	4.90
	$t_{90}$	6.996	5.4318	4.022	6.580	0.6246	0.6035	0.6574	0.5798	3.35	4.50	3.06	4.21
	$t_{80}$	6.966	5.169	4.1032	6.7109	0.6142	0.5938	0.5627	0.5673	3.40	4.76	5.93	4.75

The permeability of a small monomer molecule into a polymer membrane depends on the diffusivity as well as the solubility or sorption of the monomer in the membrane. Therefore, the sorption coefficient has been calculated using<sup>15</sup>:

$$S = \frac{M_{\infty}}{M_p} \quad (6)$$

where  $M_{\infty}$  is the mass of the solvent taken up at equilibrium swelling and  $M_p$  is the mass of the polymer sample. The values of  $S$  are given in Table 6. The sorption coefficient is found to be maximum for CV and mixed and minimum for DCP. The EV system has intermediate values. The maximum sorption coefficient of CV and mixed systems indicates that the monomer molecules are easily absorbed and accommodated in these network systems. The DCP system shows the lowest capability to accommodate the monomer molecules.

The permeabilities of natural rubber to liquid molecules can be expressed mathematically by the equation<sup>16</sup>:

$$P = DS \quad (7)$$

where  $P$  is the permeability coefficient. The permeability coefficients estimated are given in Table 6. It is noticed that the peroxide system has the highest value, highlighting the high permeability of the peroxide system.

The diffusion coefficient characterizes the ability of the monomer molecule to move among the polymer segments. The sorption coefficient is related to the equilibrium sorption of the penetrant. A large value of  $S$  shows a tendency for the monomer to dissolve into the polymer. The permeability coefficient implies the net effect of sorption and diffusion.

On analysing Table 6 we can see that the DCP system shows the highest values of diffusion coefficient and permeation coefficient, while CV and mixed systems show the lowest values. As the values of the sorption coefficient are low compared to the diffusion coefficient, it can be concluded that the permeability is controlled significantly by the process of diffusion. By analysing the effect of temperatures, it can be seen that diffusion and permeation are activated by temperature (Table 6). But the effect of temperature on sorption does not show a regular trend (Table 6).

The swelling data have been used to calculate the monomer-polymer interaction parameter,  $\chi$  and the molecular weight between crosslinks,  $M_c$ . In this approach, first, the polymer-solvent interaction parameter has been estimated by the equation<sup>18</sup>:

$$\chi = 0.34 + V_r(\delta_A - \delta_B)^2/RT \quad (8)$$

where  $V_r$  is the molar volume of solvent,  $R$  the gas constant,  $T$  the temperature, and  $\rho_A$  and  $\rho_B$  the solubility parameters of polymer and monomer, respectively. The values for the NR/styrene system at 25, 35 and 45°C are 0.801, 0.786 and 0.773, respectively. It is found that the interaction parameter changes inversely with rise in temperature. This suggests a high polymer solvent interaction at high temperature.

The molecular weight between crosslinks has been calculated from the equation<sup>10</sup>:

$$M_c = - \frac{\rho_p V \phi^{1/3}}{\ln(1 - \phi) + \phi + \chi \phi^2} \quad (9)$$

where  $V$  is the molar volume of solvent and  $\rho_p$  is the density of the polymer. The degree of crosslinking is given by<sup>19</sup>:

$$\nu = 1/(2M_c) \quad (10)$$

The values of  $M_c$  and  $\nu$  are given in Table 7. The values given in the table highlight the fact that the peroxide system has the highest crosslink density and CV and mixed systems with polysulfidic linkages have the lowest crosslink density. This again explains why the monomer uptake is highest in CV and mixed systems and lowest in the peroxide system.

The temperature dependence of transport coefficients ( $P$ ,  $D$  and  $S$ ) can be used for calculating the energy of activation for the processes of diffusion and permeation, from the Arrhenius relationship<sup>15</sup>:

$$X = X_0 \exp(-E_X/RT) \quad (11)$$

where  $X$  is  $P$ ,  $D$  or  $S$  and  $X_0$  represents  $P_0$ ,  $D_0$  or  $S_0$ , which are constants;  $E_X$  is the activation energy,  $R$  the universal gas constant and  $T$  the absolute temperature.

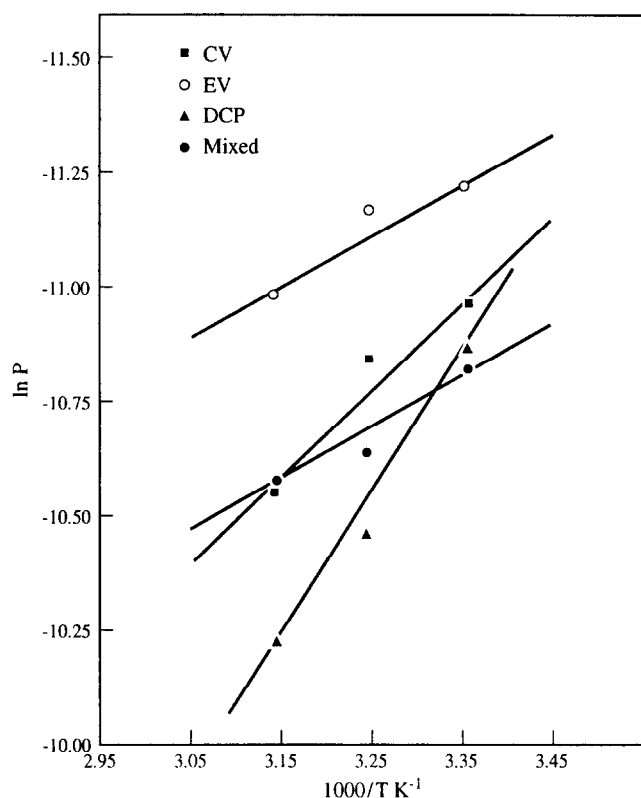
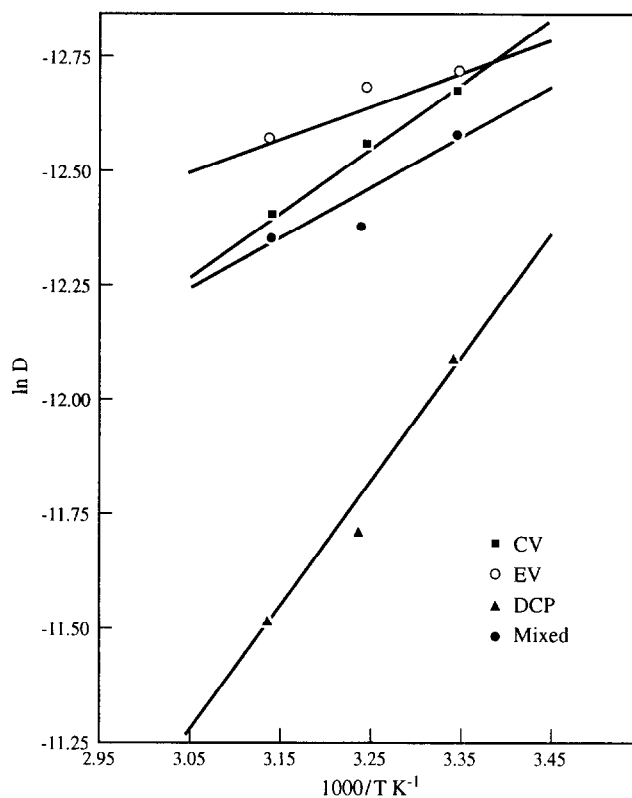
The Arrhenius plots of  $\log D$  and  $\log P$  versus  $1/T$  are shown in Figures 10 and 11, respectively. From the slopes of the curves,  $E_p$  and  $E_D$  can be calculated by linear regression analysis. The correlation coefficients are very

**Table 6** Values of diffusion coefficient, sorption coefficient and permeation coefficient at various temperatures for different curing systems

Temp. (°C)	Cure time	$D \times 10^7$ (cm <sup>2</sup> s <sup>-1</sup> )				$S$ (g/g)				$P \times 10^7$ (cm <sup>2</sup> s <sup>-1</sup> )			
		CV	EV	DCP	Mixed	CV	EV	DCP	Mixed	CV	EV	DCP	Mixed
25	$t_{100}$	2.68	3.60	5.83	2.63	6.337	4.678	3.491	6.287	16.98	16.84	20.34	16.51
	$t_{90}$	3.05	2.98	5.50	3.45	5.694	4.512	3.5172	5.8216	17.35	13.43	19.28	20.07
	$t_{80}$	3.41	2.71	5.91	3.05	5.771	4.396	3.5394	5.7381	19.67	11.89	20.82	17.47
35	$t_{100}$	3.11	3.15	6.50	3.28	6.257	4.5816	3.584	6.446	19.43	14.42	23.27	21.12
	$t_{90}$	3.60	3.11	8.16	4.20	5.6705	4.5161	3.4827	5.870	20.41	14.02	28.39	24.05
	$t_{80}$	3.68	3.73	4.65	2.96	5.6840	4.376	3.6869	6.1341	20.90	16.30	17.11	18.14
45	$t_{100}$	3.26	4.53	6.85	4.26	6.773	6.2066	3.6108	6.461	22.07	28.08	24.72	27.51
	$t_{90}$	4.11	3.45	9.93	4.28	6.3457	4.9266	3.647	5.968	26.05	16.97	36.14	25.80
	$t_{80}$	3.81	3.86	5.80	3.37	6.3181	4.6888	3.7216	6.086	24.04	18.06	21.57	20.48

**Table 7** Values of  $M_c$  and degree of crosslinking

System	Cure time	$M_c$			$\nu \times 10^{+5}$		
		25°C	35°C	45°C	25°C	35°C	45°C
CV	$t_{100}$	13208.37	13840.5	15964.96	3.785	3.612	3.131
	$t_{90}$	11684.9	12212.9	15019.18	4.279	4.094	3.329
	$t_{80}$	11834.02	12378.2	14648.96	4.225	4.039	3.413
EV	$t_{100}$	9211.52	9553.67	11029.84	5.4279	5.233	4.533
	$t_{90}$	8154.95	9462.7	11014.5	6.131	5.283	4.539
	$t_{80}$	8628.68	9288.19	10430.01	5.794	5.383	4.793
DCP	$t_{100}$	6773	7421.3	8049.8	7.382	6.737	6.2113
	$t_{90}$	6833.6	7331.3	8330.6	7.316	6.82	6.0119
	$t_{80}$	6787.7	7633.38	8008.43	7.366	6.550	6.243
Mixed	$t_{100}$	12966.2	14235.7	15325.02	3.856	3.512	3.262
	$t_{90}$	11860.7	12681.02	13990.03	4.215	3.942	3.573
	$t_{80}$	11717.2	13350.7	14041.9	4.267	3.745	3.560

**Figure 10** Arrhenius plots of  $\ln P$  vs.  $1/T$ **Figure 11** Arrhenius plots of  $\ln D$  vs.  $1/T$ 

close to 0.99. The activation energy of permeation  $E_p$  is found to be greater than the activation energy of diffusion  $E_D$ . The values of  $E_p$  and  $E_D$  are given in Table 8. The values given are for the  $t_{90}$  samples for CV, EV, DCP and mixed systems.

The heat of sorption  $\Delta H_s$  is given by:

$$\Delta H_s = E_p - E_D \quad (12)$$

The values of  $\Delta H_s$  are also given in Table 8. The enthalpy and entropy of sorption have been calculated using the van't Hoff relation<sup>20</sup>:

$$\ln K_s = \Delta S/R - \Delta H_s/RT \quad (13)$$

The values obtained by the linear regression method are given in Table 9. The values of  $\Delta S$  and  $\Delta H$  are positive.

**Table 8** Values for activation energy

System	$E_p$ (kJ mol <sup>-1</sup> )	$E_D$ (kJ mol <sup>-1</sup> )	$\Delta H_s$ (J K <sup>-1</sup> mol <sup>-1</sup> )
CV	15.970	11.76	4.21
EV	7.866	5.74	2.12
DCP	24.805	23.34	1.465
Mixed	9.958	8.564	1.394

From the values of  $\Delta H$  and  $\Delta S$  the free energy  $\Delta G$  has been calculated as:

$$\Delta G = \Delta H - T\Delta S \quad (14)$$

The  $\Delta G$  values are given in Table 9. It is interesting to note that the  $\Delta G$  values of all the systems are negative, showing spontaneous sorption.

**Table 9** Thermodynamic functions  $\Delta H$ ,  $\Delta S$  and  $\Delta G$ 

System	Cure time	$\Delta H$ (kJ mol <sup>-1</sup> )	$\Delta S$ (J K <sup>-1</sup> mol <sup>-1</sup> )	$-\Delta G$ (kJ mol <sup>-1</sup> )
CV	$t_{100}$	2.529	33.7	7.513
	$t_{90}$	4.2089	28.48	4.278
	$t_{80}$	3.518	26.2	4.278
EV	$t_{100}$	2.5866	21.42	3.796
	$t_{90}$	1.1234	16.18	3.698
	$t_{80}$	2.5622	20.60	3.5766
DCP	$t_{100}$	1.1548	14.2	3.0708
	$t_{90}$	1.4436	15.35	3.1307
	$t_{80}$	1.991	17.25	3.1495
Mixed	$t_{100}$	1.1596	15.83	3.5577
	$t_{90}$	1.076	18.24	4.359
	$t_{80}$	1.9027	21.05	4.370

## CONCLUSION

From this study on the diffusion of styrene monomer through natural rubber, it is clear that the crosslinking system has an important role in determining the diffusion coefficient. The uptake is in the order: peroxide < EV < CV, mixed. This has been explained based on the flexibility of the network and the crosslink density. The presence of rigid C–C linkages and high crosslink density of peroxide system account for the lowest monomer uptake in the peroxide-crosslinked system. On the other hand, the relatively flexible polysulfidic linkages present in the CV and mixed systems and their low crosslink density account for their high monomer uptake. As expected the EV system showed intermediate behaviour. Also, the extent of curing influences the diffusion characteristics. The uptake is lowest in  $t_{90}$  samples and it is high for both  $t_{100}$  and  $t_{80}$  samples. It is found that the temperature activates the diffusion process in all cases. The values of  $n$  are suggestive of a slight deviation from

the Fickian mode of diffusion. The polymer–solvent interaction parameter  $\chi$  was calculated and decreased with temperature. The free energy of sorption was found to be negative for all systems, suggesting spontaneous sorption.

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